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Effect of molecular weight distribution on e-beam exposure properties of polystyrene

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Abstract

Polystyrene is a negative electron beam resist whose exposure properties can be tuned simply by using different molecular weights (Mw). Most previous studies have used monodisperse polystyrene with a polydispersity index (PDI) of less than 1.1 in order to avoid any uncertainties. Here we show that despite the fact that polystyrene's sensitivity is inversely proportional to its Mw, no noticeable effect of very broad molecular weight distribution on sensitivity, contrast and achievable resolution is observed. It is thus unnecessary to use the costly monodisperse polystyrene for electron beam lithography. Since the polydispersity is unknown for general purpose polystyrene, we simulated a high PDI polystyrene by mixing in a 1:1 weight ratio two polystyrene samples with Mw of 170 and 900 kg mol⁻¹ for the high Mw range, and 2.5 and 13 kg mol⁻¹ for the low Mw range. The exposure property of the mixture resembles that of a monodisperse polystyrene with similar number averaged molecular weight (Mm, which indicates that it is Mn rather than Mw (weight averaged molecular weight) that dominates the exposure properties of polystyrene resist. This also implies that polystyrene of a certain molecular weight can be simulated by a mixture of two polystyrenes having different molecular weights.

1. Introduction

Despite its low throughput, electron beam lithography (EBL) is still the most popular nanolithography method for research and development, with its performance mainly determined by the resist for a given tool. Positive resist is more popular for EBL because of the availability of the benchmark resist poly(methyl methacrylate) (PMMA), as well as ZEP resist that offers higher sensitivity and etching resistance than PMMA. For some applications such as the fabrication of hole arrays in a film of Au, liftoff is the only option for pattern transfer since Au cannot be etched by reactive ion etching (RIE). Then negative resist would offer substantially shorter exposure time in the liftoff process, except when using a complicated 'resist tone reversal' process [1]. Bilenberg et al have selected four negative EBL resists and compared their performance: calixarene (Tokuyama Corp), ma-N 2401 (Microresist Technology), SU-8 (Microchem Corp) and mr-L 6000 (Microresist Technology) [2]. Hydrogen silsesquioxane (HSQ) is another attractive negative resist used in recent years.

However, none of the above has gained similar popularity to PMMA and ZEP resists because each has its own limit, such as the low resolution for chemically amplified SU-8 and mr-L 6000 resists, the low sensitivity of calixarene and the instability of HSQ [3].

In addition, all the above resists are commercially formulated with typically high cost and short shelf life. Therefore, it is desirable to have a negative resist like PMMA, which is a simple polymer with low cost and practically unlimited shelf life that can be dissolved easily using various solvents to give desired film thickness. Polystyrene is such a resist, as it undergoes cross-linking when exposed to deep UV light or an electron beam. Many solvents that can dissolve the linear polystyrene can be used as its developer, such as acetone, tetrahydrofuran, chlorobenzene, anisole and xylene. Polystyrene is much more resistant to plasma etching than PMMA. In addition, it can be dry thermally developed to reduce pattern collapse due to capillary force when using solvent development [4], and it can also be thermally evaporated for EBL on any (non-flat, non-planar) surface such as on an optical fiber [5]. More importantly, its exposure properties such as sensitivity and resolution can be tuned simply by using different molecular weights (Mw). Previously we have demonstrated ultra-high resolution patterning using a low Mw of 2 kg mol⁻¹ [6] and ultra-high sensitivity using a high Mw of 900 kg mol⁻¹ [7]. In addition to molecular weight, another important parameter for a polymer is the polydispersity index (PDI), which is defined as the ratio of weight averaged molecular weight \overline{Mw} and number averaged molecular weight \overline{Mn} . For a strictly monodisperse polymer, \overline{Mn} is equal to \overline{Mw} and the PDI is 1. For a polymer with broad molecular weight distribution, the PDI is much higher than 1. The PDI is known to be an important parameter for micro-phase separation of a mixture of heterogeneous polymer [8], as well as for crystallization kinetics [9].

In previous studies by us [4–7] and other research groups [10–16], in order to avoid any uncertainties, polystyrene with very narrow molecular weight distribution with PDI \leq 1.06 has been utilized. Ku's paper published in 1969 also recommended polystyrene with narrow Mw distribution for EBL in order to avoid potential pinhole formation after development [17]. In this study we will show that there is no significant effect of very broad molecular weight distribution on resist contrast and achievable resolution. We didn't observe any pinhole formation after development, either. It is thus unnecessary to use monodisperse polystyrene whose cost is more than two orders higher than the general purpose polystyrene having high PDI.

2. Experimental details

Polystyrene powder with various molecular weights and polydispersity indices (from Pressure Chemical Co. and Scientific Polymer Products Inc.) was dissolved in chlorobenzene with a concentration of 1-2 w/v%. The film thickness depends on the polystyrene concentration and spin speed. A relatively thick film is desirable for a contrast curve measured by AFM since the measurement is more accurate for thicker films, whereas a relatively thin film is preferred for high resolution line array exposure because pattern collapse due to capillary force is less severe for thinner films. Before spin-coating, the silicon wafer was cleaned using acetone and 2-propanol, followed by nitrogen drying. The films were baked at 120°C for 5 min on a hotplate for the high Mw samples (170 kg mol⁻¹ and above). For low Mw polystyrene resist (13 kg mol⁻¹ and below), we decreased the baking temperature to 70 °C and increased the time to 1 h in order to obtain a uniform and homogeneous film.

Electron beam lithography was performed using a LEO 1530 field emission SEM that is equipped with Nabity nanometer pattern generation system (NPGS) and a Raith 150^{TWO} tool. The pattern was exposed using the LEO tool at 20 kV with 100 pA beam current (20 μ m aperture) for the high Mw polystyrene samples, and using the Raith tool at 20 kV with 340 pA beam current (30 μ m aperture) for the low Mw polystyrene resists. After exposure, the samples were developed using tetrahydrofuran (THF) for 3 min for the high Mw polystyrene (170–900 kg mol⁻¹), followed by

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a 2-propanol rinse. This relatively long development time is needed since high Mw polystyrene is less soluble in the solvent than the low Mw one $(2.5-13 \text{ kg mol}^{-1})$ that was developed for 40 s.

To investigate the effect of PDI on exposure properties, since the PDI of commercially available general purpose polystyrene (expected to have unimodal Mw distribution) is unknown, we mixed two monodisperse polystyrenes with very different molecular weights (thus bimodal distribution) to mimic a polystyrene with very broad molecular weight distribution and large PDI. For the high Mw range, 170 (PDI = 1.06) and 900 kg mol⁻¹ (PDI = 1.10) polystyrene solutions having the same concentration were mixed with 1:1 volume ratio, to give a film with 1:1 weight ratio after spin-coating. For the low Mw range, 2.5 and 13 kg mol⁻¹ (both PDI = 1.06) were mixed the same way.

3. Results and discussion

For a positive resist based on chain scission such as PMMA, the dependence of resist sensitivity on molecular weight is rather weak; this is because for PMMA of longer chains, though more exposure is needed to cut it to below $\sim 10 \text{ kg mol}^{-1}$ (threshold Mw for fast dissolution in the developer), longer chains also receive proportionally more electron exposure. For instance, when Mw is decreased from 2200 to 50 kg mol⁻¹, its sensitivity increases only by 26% for room temperature development, from approximately 500 to 370 μ C cm⁻² [18]. Contrary to positive resist, Mw is fundamentally critical for chain cross-linking negative resist such as polystyrene whose sensitivity is roughly proportional to its Mw [17]. This is because less cross-linking is needed for longer polystyrene chains to render them insoluble in a solvent developer. As for resist contrast and high resolution capability, though typically a high Mw PMMA of 950 kg mol⁻¹ is used for electron beam lithography, actually a Mw as low as 15 kg mol⁻¹ can achieve very high resolution of 15 nm line-width [19]. There is no obvious relationship between high resolution capability (resist contrast) and resist molecular weight for positive resist. But for a chain cross-linking negative resist such as polystyrene, gel formation theory predicts that resist contrast increases as Mw decreases [20, 21], which is in agreement with our previous studies [6, 7]. Given such an overwhelming significance of Mw on the exposure properties of polystyrene, we studied the effect of polydispersity index over both high and low Mw ranges.

3.1. Polystyrene with high Mw

The contrast curves exposed at 20 keV for 170 kg mol⁻¹ polystyrene (PDI = 1.06), 900 kg mol⁻¹ polystyrene (PDI = 1.10) and the 1:1 mixture of the two are shown in figure 1. As expected, the sensitivity for 900 kg mol⁻¹ is much higher than that for 170 kg mol⁻¹. The sensitivity for 900 kg mol⁻¹ is comparable to typical chemically amplified resists such as SU-8; and that for 170 kg mol⁻¹ is close to ZEP-520A resist when using a strong developer [23], which is much higher than when using regular high resolution developer. For the



Figure 1. Contrast curves exposed at 20 keV for polystyrene resist of 170, 900 kg mol⁻¹ and a mixture of the two with 1:1 weight ratio. The inset shows a possible configuration of a portion of cross-linked polystyrene. (Adapted from [22], reprinted with permission. Copyright 1997 American Chemical Society.)

mixture with 1:1 weight ratio (i.e. $N_1M_1 = N_2M_2$, $M_1 = 170 \text{ kg mol}^{-1}$, $M_2 = 900 \text{ kg mol}^{-1}$), if we assume the two components are strictly monodisperse (i.e. PDI = 1.0), the weight averaged molecular weight can be calculated as $\overline{Mw} = (N_1M_{12} + N_2M_{22})/(N_1M_1 + N_2M_2) = (170 + 900)/2 = 535 \text{ kg mol}^{-1}$; for a 1:1 weight ratio, the number ratio is $N_1:N_2 = 900:170$, thus the number averaged molecular weight can be calculated as $\overline{Mn} = (170 \times 900 + 900 \times 170)/(170 + 900) = 286 \text{ kg mol}^{-1}$. By definition, the polydispersity index of the mixture is PDI = 535/286 = 1.87. The calculated contrast for the mixture ($\gamma = 1.3$) lies between those of the two components ($\gamma = 0.9$ and 1.9 for

900 and 170 kg mol⁻¹, respectively). As can be seen from the figure, the contrast curve for the 1:1 mixture resembles a hypothetical monodisperse polystyrene resist with Mw \sim 300 kg mol⁻¹. Therefore polystyrene of certain molecular weight can be simulated by a mixture of two polystyrenes having very different Mw. It is thus unnecessary to have a large number of polystyrene resists with different Mw for different applications using EBL.

To study the resolution capability of the polystyrene samples, we exposed dense line arrays each over an area of $(5 \,\mu m)^2$ that is larger than the range of electron backscattering for 20 keV, so a similar dense array is expected when patterning over larger area. The lines were exposed as single pass lines with step size of 5 nm. As seen in figure 2, the 150 nm period line array pattern was better defined using the 1:1 mixture than 900 kg mol⁻¹. The lower Mw of 170 kg mol^{-1} can well define a line array of 80 nm period. For comparison, we also studied the resolution capability for a general purpose 260 kg mol^{-1} polystyrene with high polydispersity index (figure 2d), which showed improved pattern definition with smoother lines than the 170 kg mol⁻¹ having narrow molecular weight distribution (see below for an explanation). This again manifests that the resist performance greatly depends on molecular weight, but not on its distribution.

3.2. Polystyrene with low Mw

The contrast curves exposed at 20 keV for 2.5 kg mol⁻¹ polystyrene (PDI = 1.06), 13 kg mol⁻¹ polystyrene (PDI = 1.06) and the 1:1 mixture of the two are shown in figure 3. As expected, they all showed considerably higher contrast than



Figure 2. Dense line arrays exposed at 20 keV in polystyrene with different molecular weights: (a) Monodisperse 900 kg mol⁻¹, 150 nm period, 0.26 nC cm⁻¹ line dose; (b) 1:1 mixture of 170 and 900 kg mol⁻¹, 150 nm period, 0.66 nC cm⁻¹; (c) monodisperse 170 kg mol⁻¹, 80 nm period, 1.0 nC cm⁻¹; (d) general purpose polydisperse 260 kg mol⁻¹, 80 nm period, 0.91 nC cm⁻¹.

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Mw (kg mol ⁻¹)	Mn (kg mol ⁻¹)	Contrast	Sensitivity D_{50} (μ C cm ⁻²)	Gel point D ₀ (μ C cm ⁻²)	$\mathrm{D}_{50}\times\mathrm{Mw}$	$\mathrm{D}_{50}\times\mathrm{Mn}$	$D_0 imes Mw$	$D_0 imes Mn$
2.5	2.4	3.6	3400	2310	8 500	8160	5775	5544
7.8	4.2	3.5	1750	1220	13 650	7350	9516	5124
13	12.3	3.1	600	370	7 800	7380	4810	4551

Table 1. Comparison of exposure properties of the three polystyrene resists, with the values derived from figure 3.



Figure 3. Contrast curves exposed at 20 keV for polystyrene resists of 2.5 and 13 kg mol⁻¹ and a mixture of the two with 1:1 weight ratio. The definition of gel point dose D₀ and sensitivity D₅₀ is indicated.

the high Mw polystyrenes studied above, at the cost of greatly reduced sensitivity. For the 1:1 weight ratio mixture of 2.5 and 13 kg mol⁻¹, \overline{Mw} , \overline{Mn} and PDI is calculated as 7.75 kg mol⁻¹, 4.19 kg mol⁻¹ and 1.85, respectively. The contrast curve for the 1:1 mixture resembles a hypothetical monodisperse polystyrene resist with a Mw of $4-5 \text{ kg mol}^{-1}$. Since for low Mw polystyrene the resist surface after development is much smoother than for high Mw polystyrene (this is related to its higher resolution capability), the contrast curve measured by AFM is more accurate than that for high Mw polystyrene resists that gave rough resist surfaces after development. It is thus more meaningful to carry out a quantitative study with the low Mw resists. Table 1 lists the three resists' gel point dose D₀, sensitivity D₅₀ (dose for 50% normalized thickness), contrast, as well as the four products between $M_{w,n}$ and $D_{0.50}$. Theoretically the dose-molecular weight product should be independent of molecular weight. Apparently the products of dose and number averaged molecular weight are similar for the three resists, but those of dose and weight averaged molecular weight are very different. Moreover, Mn for the mixture is closer to that of 2.5 kg mol⁻¹ than 13 kg mol⁻¹, so is the resist contrast. We can therefore conclude that the resist properties for polystyrene depend mainly on \overline{Mn} if it differs from \overline{Mw} . It is known that for a polymer with broad molecular weight distribution, some properties such as tensile strength depend mainly on \overline{Mn} , whereas other properties such as viscosity depend mainly on Mw [24, 25]. Longer chains are more important for the value of \overline{Mw} , whereas shorter ones with larger number are more important for the value of Mn. Therefore, here the exposure property is mainly determined by the low Mw component due to its large number.

been studied previously. Feit et al showed that for the same molecular weight of 233 kg mol⁻¹, D_{80} (dose for 80% normalized thickness) for polystyrene with higher PDI is considerably higher than near-monodisperse polystyrene, though the difference for D_{50} (dose for 50% normalized thickness) is less [26]. They thus concluded that PDI is a very important parameter for a polystyrene resist. We disagree with their conclusion, and believe that a higher dose was needed in their study for the resist with larger PDI because its \overline{Mn} is lower for the same \overline{Mw} . Lai *et al* claimed that the contrast of polystyrene resist decreases with an increase in molecular weight distribution [27]. Yet this conclusion was drawn based on the contrast curves for several polystyrene resists that differed not only in PDI but also in molecular weight. Gel formation theory also implies higher contrast for uniform Mw distribution (PDI = 1) than random Mw distribution (PDI = 2) [20]. However, it was assumed that gel fraction is equal to normalized remaining film thickness after development, which is inaccurate since gel formation is a 'bulk' process occurring throughout the film thickness whereas resist dissolution by a developer is a layer by layer surface process. Our results indicate that PDI has an insignificant effect on resist performance. One plausible explanation for this is that once two polymer chains have been cross-linked by electron beam exposure to form one larger chain, the chain length of the two original polymer molecules is no longer important, be it two molecules of equal length or not.

Polystyrene resists with different Mw and PDI have

Lastly, we performed gel permeation chromatography (GPC) measurement on the above-mentioned general purpose polystyrene with a molecular weight of 260 kg mol⁻¹, which showed $\overline{Mw} = 286$ kg mol⁻¹, $\overline{Mn} = 64$ kg mol⁻¹, and PDI = 4.5. Due to its lower \overline{Mn} , this polymer achieved better pattern definition than the monodisperse (PDI = 1.06) 170 kg mol⁻¹ one as seen in figure 2. We also compared it with a monodisperse polystyrene of 62 kg mol⁻¹ (PDI = 1.06, thus $\overline{Mn} = 58$ kg mol⁻¹) that is commercially available. As expected, and shown in figure 4, though their \overline{Mw} differs by 4.6 times, the sensitivity of the two polymers is reasonably close (90 and 115 μ C cm⁻², respectively). This again showed that number averaged molecular weight dominated the exposure properties of polystyrene resist.

4. Conclusions

Despite the fact that polystyrene's sensitivity is inversely proportional to its Mw, no significant effect of very broad molecular weight distribution on the sensitivity, contrast and achievable resolution is observed. It is thus unnecessary



Figure 4. Contrast curves exposed at 20 keV for monodisperse (PDI = 1.06) and general purpose polydisperse (PDI = 4.5) polystyrene with a similar number averaged molecular weight of $\sim 60 \text{ kg mol}^{-1}$.

to use the costly monodisperse polystyrene for EBL. In our approach, we simulated a polystyrene with high PDI by mixing at 1:1 weight ratio two polystyrenes with very different molecular weights. The exposure property of the mixture resembles that of a monodisperse polystyrene with a close number averaged molecular weight \overline{Mn} , which indicates that it is \overline{Mn} rather than \overline{Mw} (weight averaged molecular weight) that dominates the exposure properties of polystyrene resist. We also studied a general purpose polystyrene with a high polydispersity of 4.5, and found that its properties is comparable to those of monodisperse polystyrene having similar number averaged molecular weight. Therefore, one can use the low-cost general purpose polystyrene for EBL; and one doesn't need to have a large number of polystyrene resists with different molecular weights for different applications using EBL, as any given molecular weight can be simulated by a mixture of two polystyrene samples having different molecular weights.

References

- Hajiaboli A, Cui B, Kahrizi M and Truong V V 2009 Phys. Status Solidi a 206 976
- [2] Bilenberg B, Schøler M, Shi P, Schmidt M S, Bøggild P, Fink M, Schuster C, Reuther F, Gruetzner C and Kristensen A 2006 J. Vac. Sci. Technol. B 24 1776
- [3] Clark N, Vanderslice A, Grove R and Krchnavek R R 2006 J. Vac. Sci. Technol. B 24 3073
- [4] Con C, Abbas A S and Cui B Presented at EIPBN May 2012
- [5] Zhang J and Cui B Presented at EIPBN May 2012
- [6] Ma S, Con C, Yavuz M and Cui B 2011 Nanoscale Res. Lett. 6 446
- [7] Con C, Dey R, Ferguson M, Zhang J, Mansour R, Yavuz M and Cui B 2012 *Microelectron. Eng.* 98 254
- [8] Liping L, Yajiang H and Qi Y 2011 J. Macromol. Sci. B 50 2140
- [9] Chen C, Hsieh T and Ju M 2009 J. Alloys Compounds 480 658
- [10] Itaya K, Shibayama K and Fujimoto T 1982 J. Electrochem. Soc. 129 663
- [11] Manako S, Fujiga J, Ochiai Y, Nomura E and Matsui S 1997 Japan. J. Appl. Phys. 36 7773
- [12] Manako S, Fujiga J, Ochiai Y, Nomura E and Matsui S 1997 Japan. J. Appl. Phys. 36 L724
- [13] Ochiai Y, Manako S, Fujita J and Nomura E 1999 J. Vac. Sci. Technol. B 17 933
- [14] Austin M, Zhang W, Ge H, Wasserman D, Lyon S and Chou S Y 2005 Nanotechnology 16 1058
- [15] Whipps P W 1979 Microcirc. Eng. 118
- [16] Jagt J C and Whipps P W 1980 Philips Tech. Rev. 39 346
- [17] Ku H Y and Scala L C 1969 J. Electrochem. Soc.: Solid State Sci. 116 980
- [18] Yan M, Choi S, Subramanian K R V and Adesida I 2008 J. Vac. Sci. Technol. B 26 2306
- [19] Khoury M and Ferry D K 1996 J. Vac. Sci. Technol. B 14 75
- [20] Atoda N and Kawakatsu H 1976 J. Electrochem. Soc.: Solid State Sci. Technol. 123 1519
- [21] Sugita K and Ueno N 1992 Prog. Polym. Sci. 17 319
- [22] Sheiko S S, Gauthier M and Möller M 1997 Macromolecules 30 2343
- [23] Shokouhi B, Zhang J and Cui B 2011 Micro & Nano Lett. 6 992
- [24] McCormick H W, Brower F M and Kin L J 1959 Polym. Sci. 39 87
- [25] Merz E H, Nielson L E and Buchdahl R 1951 Indust. Eng. Chem. 43 1396
- [26] Feit E and Stillwagon L 1980 Polym. Eng. Sci. 20 1058
- [27] Lai J H and Shepherd L T 1979 J. Electrochem. Soc. 126 696